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# Thermal Stability of Foils Made of Graphene-Oxide and Grapheme -Oxide with Fullerene and their Composites with Methyl Car Boxy Cell u lose and with Beta 1, 3/1, 6 – D-Glucan

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*Abstract-* This contribution contains data on thermal stability of certain materials whose initial precursor is graphite. Graphite was oxidized separately and in a mixture with fullerene  $C_{60}$ . The prepared oxides were processed with vacuum filtration to produce foils and their morphology and thermal stability was described. The graphene oxides reacted with nano-cellulose and  $\beta$ - glucan to produce composites. The prepared composites in the form of foils were tested for thermal stability and further analyzed e.g. by FT-IR, SEM, etc.

Keywords: grapheme oxide, fullerene -  $c_{\omega}$  Intercalate, composite, nano-cellulose,  $\beta$ -glucan. GJRE-H Classification : FOR Code: 091099



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# Thermal Stability of Foils Made of Graphene-Oxide and Grapheme -Oxide with Fullerene and their Composites with Methyl Car Boxy Cell u lose and with Beta 1, 3/1, 6 – D-Glucan

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Abstract- This contribution contains data on thermal stability of certain materials whose initial precursor is graphite. Graphite was oxidized separately and in a mixture with fullerene  $C_{60}$ . The prepared oxides were processed with vacuum filtration to produce foils and their morphology and thermal stability was described. The graphene oxides reacted with nano-cellulose and  $\beta$ - glucan to produce composites. The prepared composites in the form of foils were tested for thermal stability and further analyzed e.g.by FT-IR, SEM, etc.

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### I. INTRODUCTION

Graphite is an allotropic modification of carbon with sp<sup>2</sup> bonds and made up of layers of mutually interconnected hexagonal rings. The layers are arranged in parallel planes 335 pm apart. Carbon atoms in the adjoining layers are not chemically bonded to each other and they are attached by weak van der Waals forces that make it possible for various atoms or molecules in liquid or gaseous form to get in between the carbon layers. The resulting substances are called intercalation compounds of graphite and their characteristic parameter is the so-called "degree of intercalation", which indicates the number of carbon layers between two layers of an inter calated substance (Klouda, 1985).

Depending on a type of the intercalated substance the graphite plane may be either an acceptor or donor of electrons. Another option is the so-called  $\pi$ -complex created by intercalation of substances of AX<sub>y</sub> type, where A is a metal or non-metal with a high valence status, X is an electronegative element and y is a stoichio metric coefficient.

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Intercalates of graphite with alkali metals have been known since 1930s. They are called intercalates of the first degree with the formula  $C_8M$  (M=K, Rb, Cs), i.e. they are characterized by a stacking sequence of layers of carbon and alkali metal.

Intercalates of graphite with alkali metals or in combination with other metals have been used in a number of applications as catalysts, e.g. for synthesis of ammonia, synthesis of carbohydrates by hydrogenation of carbon oxides, hydrogenation of olefins, they have sorption properties etc. (Klouda, 1985). Substituents can be chemically bonded to graphite under certain conditions by fluorination or oxidization.

Fluorination of graphite with elemental fluorine at 400-600°C produces a covalent compound called fluoro graphite CF<sub>x</sub>, x=0.25-1.12, depending on reaction conditions of the fluorination (Klouda, 1985). Oxidization of graphite with strong oxidizing agents produces grapheme oxide (GO), which is a precursor for chemical preparation of graphene (Makharza et. al., 2013).

Publications dealing with oxidization of graphite to prepare G-O usually specifya method use das described by specific authors: Hoffmann (HNO<sub>3</sub>, KCIO<sub>3</sub>), Stay dennaier (HNO<sub>3</sub>, KCIO<sub>3</sub>), Tour (P<sub>2</sub>O<sub>5</sub>, KMnO<sub>4</sub>), Hummers (NaNO<sub>3</sub>, KMnO<sub>4</sub>). In all those methods the mainchemical agent used is concentrated sulfuric acid (Chang and Pumera, 2013).

GO is a compound made up of a carbon skeleton with main functional groups, such as carboxyl, carbonyl, epoxy and ether group sandhydroxy-groups. These functional group senable chemical Ireactions of GO (Zang et. al., 2011) to form covalent bonds with other compounds (e.g.esterification, amidation).

Another option is a GO reaction to form noncovalent bonds (Makharza et. al., 2013). The possible types of the bonds are hydrogen bonds, van der Waals forces, H- $\pi$ , cation - $\pi$ , anion- $\pi$ ,  $\pi$ - $\pi$ , electrostatic forces. These non-covalent bonds are employed in preparation of composite polymers, biopolymers (Yoo, B. M. et. al., 2013) and in use of GO adsorption and absorption properties (Kyzas et. al., 2014; Fakhri et al., 2013; Chabot et. al., 2014).GO suspension can be vacuum filtered to prepare foils that find use in biology, electrical

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engineering, optics (Russo et. al., 2013) and biomedicine (She n et. al., 2012).

Graphene can be prepared by a chemical method which consists in reduction of oxidized carbon (functional groups) in GO with various reducing agents (hydrazine, metal hydride, hydrogen, hydrogen iodide) or reducing methods, such as reflux in a polar solvent, microwaves irradiation, electrochemical reduction (Dreyer et. al., 2010).

The composition of graphene-oxide and its decomposition by an exothermic creaction in dicates a potential fire risk. This shall apply mainly to industrial production, processing and storage of GO-materials. If stored in a solid form GO shall be protected against sources of heat, electric discharges and exposure to high-intensity light. A question which appeared in publications by Krishnan et al. (Krishnan et al., 2012), i.e. whether GO is afire retarder or fire hazardous material, has defined the objective of this work in which we want to assess behavior of GO and GO-C<sub>60</sub> foils and their composites with cellulose and  $\beta$ -glucan when thermally exposed.

### a) Experimental Part

### i. Employed chemicals

Graphite PM – very fine cry stall in epowder graphite, mesh0.025mm, Supplier: Koh-I-Noor Netolice, Czech Republic

Fullerene C $_{\rm 60},~99.5\% purity,~SES$  Research, Houston USA

Sulfuric acid, nitric acid, potassium permanganate, sodium hydro carbonate, CM –cellulose C4146 – Supplier: Sigma –Aldrich

Beta 1, 3/1, 6 – D – glucan (59% beta, 9% alpha), botanic source oyster mushroom, Supplier: Dimenzia s.r.o., Kežmarok Slovakia.

#### ii. Employed methods

Ultrasonification with PS4000A, power output 500W, thermostat 0-77°C, frequency 35 kHz

ATR analysis by means of FTIR spectrometry was performed using the spectrometer Brucker Aplha/FT-IR, ART crystal (identified as Platinum Diamond 1 Ref1), software OPUS 6, 5, source IR SiC Globar. The number of spectrum scans was 24, resolution 4 cm-<sup>1</sup>, spectrumrange 375-4000 cm-<sup>1</sup>.

Thermal analyses TGA and DSC of the prepared an of ibers were performed on STA 1500, Instrument Specialists Incorporated-THASS, analytical scale SUMMIT, SI 234-4, at flowrate 20 ml/min., heatingrate 10°C/min., ceramiccrucible, diameter 5 mm and height 8 mm, degradation medium: air.

Morphology of then an of ibers was determined with SEM Phenom FEI and SEM FEI Quanta 650 FEG (USA).

# b) Preparation of Graphene-Oxide (GO) and Graphene-Oxide + $C_{60}(GO-C_{60})$

Graphite was oxidized with a mixture of  $H_2SO_4$ , KMnO<sub>4</sub> and NaNO<sub>3</sub> according to Hummers and Offer man (Hummers and Offer man, 1958). Graphite, sulfuric acid and sodium nitrate (for experiments I-II also fullerene C<sub>60</sub>), were placed into a flask, the mixture was stirred and cooled to 10°C. Potassium permanganate was subsequently added into the reaction mixture through a hopper in small doses. The mixture with the permanganate was slowly heated to 60°C and stirred at that emperature for 3 hours. Then it was left to stand for three days at the laboratory temperature.

The obtained product was filtered off, washed with big quantity of distilled water until negative reaction to sulfate anions and dried for three days on a Petridish at 50-60°Ctoform foils of GO or  $\text{GO-C}_{60}$ .

Samples weights for the individual experiments I –III

- I and II: 1 g graphite PM; 35 ml  $H_2SO_4;$  2.11 g  $NaNO_3;$  0.5 g  $C_{60}$  and 4.6 g KMnO\_4
- III.: 2 g graphite PM; 45 ml  $H_2SO_4;$  2.8 g  $NaNO_3$  and6.5 g  $KMnO_4$

## c) Oxidation of Fullerene Alone $C_{60}$ (Blank Test)

In order to confirm or to refute our the oretical assumptions about behavior of fullerene in an oxidation mixture we have performed an experiment in which we maintained mutual ratios of carbon to the other reagents as those used in the experiment with graphite (0.7g C<sub>60</sub>, 1 g NaNO<sub>3</sub>, 2 g KMnO<sub>4</sub>, 18 ml H<sub>2</sub>SO<sub>4</sub>). Also there action times and subsequent treatment were equivalent. After vacuum filtration we did not obtain foils but after the drying we obtained black loose powder (hereinafterC<sub>60</sub>- oxi). The powder was investigated with FT-IR, TGA, DSC analysis and the results were compared with analyses of the initial fullerene.

# d) Reaction of GO and $GO-C_{60}$ Foils with Cellulose in Acid Environment

GO (0.3 g) and GO-C<sub>60</sub> (0.3 g) foils were placed into Erlenmeyer flasks and 10 ml of distilled water was added. Foil schanged into suspension safter 3 day so firregularstirring and short-termultrasonification. Subsequently, cellulose was added into the flasks (0.65 g) and 8 ml H<sub>2</sub> SO<sub>4</sub> (96%). The reaction was exothermic. The mixture was ultrasonificated in a water bath at 40°C. Then the flask content was poured into 50 ml of distilled water and neutralized with a solution of sodium hydro carbonate (NaHCO<sub>3</sub>) until neutral reaction. The product was then vacuum filtered and washed on a filter with ca. 40 ml of distilled water, dried at 50°C on a Petri dish on which it formed foils.

### e) Reaction of GO with B-Glucan

Graphene oxide (prepared according to Hummers, 1958) in the form off oils (0.3g) was placed

into Erlenmeyer flasks with 25 ml of distilled water. No suspension was formed after 48 hours and it was necessary to performer peated ultrasonification 5x 2 minutes to prepare the suspension. Beta – glucan (0.45 g) was added into the suspensions and in one of the flasks also 0.7 ml of concentrated  $H_2SO_4$  (an exothermic process occurred after the acid was added).

- i. Visualization of the flask content
- without H<sub>2</sub>SO<sub>4</sub> gel (product I)
- with H<sub>2</sub>SO<sub>4</sub> suspension(product II)

Subsequently, Erlenmeyer flasks were placed into a water bath (40°C) and ultrasonificated 3 times for 2 minutes. Then they were left to stand for 15 day sat the laboratory temperature. The flask with the product I (GO- $\beta$ G) was vacuum filtered and the resulting black foil was dried at 50-55°C. The flask with the product II

(GO-ß G, H+) was poured into a 300 ml flask, diluted with 200 ml of distilled H2O and subsequently 12 times decanted top H 6.5. The flask contained brown spongycoagu late in1/3 of the flask volume. A part of the coagulate was vacuum filtered and it formed a brown foil which was dried at 50-55°C.

## II. Results

### a) Oxidation of Graphite and Mixture of Graphite- $C_{60}$

The foil sobtained by vacuum filtration from the product of oxidation of graphite alone and the product of joint oxidation with  $C_{60}$  had clearly different morphologies (see Fig. 1). GO- $C_{60}$  foils looked more compact than GO foils. When inspecting the morphology with electron micros copy the GO- $C_{60}$  foil has a rougher surface (see Fig.2). For GO- $C_{60}$  foils we also determined its texture.





a) foil obtained by oxidation of graphite: front and back side of the foil, scale 1 mm

b) foil obtained by oxidation of graphite with fullerene  $C_{60}$ : front and back side of the foil, scale 1 mm



THERMAL STABILITY OF FOILS MADE OF GRAPHENE- OXIDE AND GRAPHEME -OXIDE WITH FULLERENE AND THEIR COMPOSITES WITH METHYL CAR BOXY CELL U LOSE AND WITH BETA 1, 3/1, 6 – D-GLUCAN



a) GO: different views



b) GO-C<sub>60</sub>: different views

*Figure 2*: Fine morphology of the foils after vacuum filtration of GO (a) and  $GO-C_{60}$  (b)

The specific surface of the foil samples GO-C<sub>60</sub> 21.9m<sup>2</sup>/g and the volume of adsorbed was monomolecular layer was 5.03 ml/g. The volume of me so pores or macro pores was 0.286 ml/gin comparison with the volume of micro pores which was0.001 ml/g. The volume representation of me so pores was ca. 286 times higher than that of micro pores. The sample had a me spoor us character with some representation of macro pores. The sample contained very little micro pores (see the volume representation - only0.001 ml/g)consisting of the pores with the diameter 0-1 nm (ca. 63%) - only one fraction of the pores, while no other micro pore fractions were identified -they were probably clogged. Theme sopores included the following fractions: 1.5-3 nm (ca. 38%), 3-5 nm (ca. 13%) and 5-10 nm (ca.5%) and 10-50 nm (ca.8%). As for macro pores, the sample contained only one fraction, while ca. 24% of the specific surface was formed by macro pores with the diameter 50-200 nm. The measured parameters are shown in Table No. 1.

*Table 1 :* Texture parameters of  $GO-C_{60}$  samples

Sample identification	a[m²/g]	B[cm³/g]	c[cm³/g]	d[cm³/g]	e[cm³/g]
$GO-C_{60}$	21.9	5.03	0.040	0.001	0.286

a – specific surface, b – volume of adsorbed monomolecular layer, c –cumul ative volume of pores, d –cumul ative volume of micro pores, e –cumulative volume of me so- and macro pores

Subsequently, the foils were examined with X-ray analysis (Fig.3), FT-IR, TGA and DSC analyses. The X-ray analysis has proved a negligible difference in expansion of the space between the layers, see Fig.3.



*Figure 3*: X-ray analysis, GO: d=739 pm, 299 pm; GO-C<sub>60</sub>:d= 718 pm, 425 pm

b) IR Spectrometry of GO and  $GO - C_{60}$ 



*Figure 4* : IR spectrum of products of oxidation of graphite and graphite with  $C_{60}$  : a) GO, b) GO- $C_{60}$ , c) comparison of the spectrums a) and b

When preparing GO by oxidation methods FT-IR is usually indicated as a method for identification of the basic functional groups. In most cases the authors indicate vibration ranges for the given groups:

3000 – 3700 cm<sup>-1</sup>v (-COOH, -OH, H2O)

 $1850 - 1750 \text{ cm}^{-1} v (-C=O)$ 

 $1650 - 1750 \text{ cm}^{-1} v$  (car boxy COOH)

 $1500 - 1600 \text{ cm}^{-1} \text{v} (\text{sp2 C}=\text{C})$ 

1280 – 1320 cm-<sup>1</sup>v (epoxides C-O-C)

At the same time, ranges that follow may contain the following vibrations:

1280 – 1500 cm-<sup>1</sup>: ethers, epoxides, ketones, peroxides, benzoquinone

1100 – 1280 cm-<sup>1</sup>: peroxides, ethers, ketones, lactones, anhydrides, epoxides, benzoquinones

 $900 - 1100 \text{ cm}^{-1}$ : lactones, peroxides, hydroxyls, 1, 3 dioxane, anhydrides, epoxides, car boxy les-OH bond vibration at 3420 cm<sup>-1</sup>, C=O bond vibrationat1720 – 1740 cm<sup>-1</sup>.

The mutual comparison of spectrums we obtained for GO and GO-C $_{\rm 60}$  is shown in Fig.4

The difference between the GO and GO-C60spectrums is in the ratio of the mutual adsorbances for the vibrations:

GO	1390	1274	1228 cm-1
GO-C <sub>60</sub>	1378	1278	1228 cm-1

In the spectrum range 700 – 450 cm  $^{-1}$  the ads or bance of GO has a medium value while for GO-C\_{60} it is

high; in general, GO-C<sub>60</sub> demonstrates higher adsorbances throughout the entire spectrum range.

#### c) Thermal Tests of GO and GO-C<sub>60</sub> foils

Two prominent peaks were detected for samples of GO and  $\text{GO-C}_{60}$  foil son the DSC curve (Fig.5) which corresponded to exothermic processes. The first exothermic process was accompanied by a substantial drop of weight: for go by 43.6% and for  $\text{GO-C}_{60}$  it was even higher-51.1% (see Tab. 2).

In the case of GO the first exothermic process occurred at 190.9°Cwith the maximum at 225°C and thermal effect of 508.4 kJ/kg; in the case of GO-C<sub>60</sub> the process started earlier, at 182.6°Cwith the maximum at 205°Cand the thermal effect was lower than for GO. The values of thermal effects in the individual temperature intervals are provided in Tab.3.The second exothermic effect in the case of GO has its maximum at 450°C, while for GO-C<sub>60</sub> it was already at 390°C and the detailed shape of the curve was different (compare Figures5 and 6). For GO-C<sub>60</sub> the weight loss during the second exothermic effect in the case of GO, which means a situation different from the first effect.

The weight losses until the first exothermic process are essentially the same for both the foils (ca. 20%) with mild end other mic effects, with a higher thermal effect for GO (anticipated dehydration). Also the overall thermal effect of decomposition is higher for GO-C<sub>60</sub> foil by ca. 30% (see Tab.3).



Figure 5: Thermal analysis of a product of graphite oxidation- foil

(degradation medium: air, air flow : 20ml/min, temperature interval: 25-600°C, heating rate 10°/min, sample weight 10.0mg)



Figure 6: Thermal analysis of a product of graphite and fullerene oxidation-foils

(degradation medium: air, air flow:20ml/min, temperature interval: 25-600°C,heating rate 10°/min, sample weight 10.0mg)

When performing the experiments we had some expectations about the course and results of there action. The functional groups expected on GO were the following: carboxyl, carbonyl, epoxide, hydroxyl and partly also lactone or sulfonic group. Fullerene  $C_{60}$  was expected to have the following groups on its

molecule: - SO<sub>3</sub>H, -OH, -NO<sub>2</sub>, -ONO<sub>2</sub>, epoxide group. Covalent bonds may form between those groups (esterification, dehydration, addition), as well as noncovalent ones – hydrogen bonds,  $\pi$ -  $\pi$  interaction, vander Waals forces. Also intercalation of a fullerene molecule may occur in the graphene-oxide space. Even breakage of a fullerene molecule in the conditions of oxidation cannot be excluded.

Table 2 :	Division	of the TGA	curve into	temperature ir	itervals

Sample No.	Interval No.	Temperature range (°C)	Weight loss (%)
	1	25.0 - 142.4	11.0
	2	142.4 – 213.5	8.8
CO faila	3	213.5 – 222.3	43.6
GO IOIIS	4	222.3 - 368.8	2.8
	5	368.8 - 473.0	18.1
	6	473.0 - 600.0	6.1
$GO-C_{60}$ foils	1	25.0 – 87.1	4.4
	2	87.1 – 153.0	8.2
	3	153.0 – 197.0	8.0
	4	197.0 – 205.0	51.1
	5	205.0 - 281.3	10.8
	6	281.3 - 490.9	18.0

Table 3 : Parameters of the ongoing thermal processes (DSC)

Sample No.	Thermal process No.	Temperature range (°C)	ΔH (kJ/kg) *	H <sub>f1</sub> (mW)	ΣΔH (kj/kg)
	1	25.0 – 154.1	874.6	15.8	
GO foils	2	190.9 – 241.1	-508.4	107.4	-910.6
	3	356.5 - 492.1	-1277.1	31.0	
	1	42.0 - 124.2	141.7	6.4	
GO-C <sub>60</sub> foils	2	182.6 – 221.5	-308.7	71.1	-1204.1
	3	319.7 – 481.6	-1037.1	28.1	

 $^{*}\Delta H =$  thermal effect of the process based on the DSC curves

 $(\Delta H > 0...endothermic process, \Delta H < 0...exothermic process)$ 

One quoted work (Trzaskowki et al., 2013) deals with energy options of a combination of graphene with  $C_{60}$  on the condition that graphene surface either defect-free or with defects. Chemical attachment of  $C_{60}$  on amonolayer of graphene is not possible for energy reasons. However, in presence of various defects on the grapheme C- layer, such as e.g. formation of 4- and 5-atomic rings, Stone-Wales defect and other types of defects (e.g. flower defect, octa-penta-, hepta-disrupted cyclic formations) may provide spaceorre active points for potential bonds with fullerene  $C_{60}$ .

One of the defects mentioned for the graphene structure is the so-called "adatom"–an adsorbed atom. It is a defect in which e.g. transition metal is adsorbedon the C lattice. Adsorption of transition metals changes physicochemical properties and biocompatibility of graphene or GO (Faye, 2012; Ne to, 2009). We have identified a similar defect in the prepared GO: it was e.g. an atom of Fe identified by EDAX in the GO structure (Fig.7).





We performed the so-called blank test to get some notion of how fullerene behaves in the oxidizing environment to which it is exposed jointly with graphite.

# d) FT-IR Analysis of $C_{60}$ -Oxi and Initial $C_{60}$

The measured IR spectrum of the initial fullerene  $C_{60}$  is provided in Fig.8. The main characteristic vibrations for  $C_{60}$  (Saeedfar et al., 2013) are 522 cm<sup>-1</sup>, 573 cm<sup>-1</sup>, 1159 cm<sup>-1</sup>, and 1426 cm<sup>-1</sup> and they are also a part of the  $C_{60}$ -oxi spectrum.

For C<sub>60</sub>-oxi (Fig.9) we have identified an additional weak vibration in the absorption interval 1556-1644 cm<sup>-1</sup> (probably C=C, C=O), 1000-1100 cm<sup>-1</sup> (probably C-O-C epoxy, alkoxy, C-CO-C) and a medium vibrationat 702 cm<sup>-1</sup> (probably a substituted romatic ring). Based on characteristic vibrations of functional groups we can probably exclude presence of the following functional groups on the C<sub>60</sub>-oxi molecule:-OH, -COOH, -COOR, -SO<sub>3</sub>H, -NO<sub>x</sub>.



Figure 9: IR-spectrum of C<sub>60</sub>-oxi

e) Thermal Stability of  $C_{60}$ -Oxi and Initial  $C_{60}$ 

In the measure demperature range, as indicated by the TGA and DSC curves (Fig.10-11) up to ca. 420°C, there were no thermal effects on either of the tested sample sand the weight loss of both the samples was comparable in units of percents (6.1% for C<sub>60</sub> and 8.6% for C<sub>60</sub>-oxi). In the temperature range of 420-600°C both the samples underwent exother micreactions, however the thermal effects were very different. The thermal effect of C<sub>60</sub>-oxi was 20 times higher than that of the initial C<sub>60</sub> (1947 kJ/kgfor C<sub>60</sub>-oxi and 180 kJ/kg for

 $C_{\rm 60}).$  The weight loss of  $C_{\rm 60}\text{-}oxi$  (56.7%) was twice bigger than that of  $C_{\rm 60}$  (28.8% ).



*Figure 10 :* Thermal analysis of C<sub>60</sub>(degradation medium: air, air flow 20 ml/min, temperature interval 25-600°C, heating rate10°/min, sample weight11.21 mg)



*Figure 11 :* Thermal analysis of C<sub>60</sub>-oxi (degradation medium: air, air flow 20 ml/min, temperature interval 25-600°C, heating rate10°/min, sample weight 11.56 mg)

The resistance of  $C_{60}$ -oxi against thermal exposure was partly disrupted. We anticipate partial oxidation of the C-skeleton (IR-spectrum Fig.9) and thus disruption of its consistence (electron balance) which means that defects can appear in the carbon structure when the material is heated -e.g. the Stone-Wales defect (Kabiretal., 2011).

We have identified a similar reduction of resistance of the fullerene carbon skeletoninits bromoderivative. After end other micdisruption of C-Brbond the  $C_{60}$  moleculefully (100%) decomposed at 420-550°C. We are fully aware of the fact that this so-called" blind test" may not completely correspond to the oxidation process in presence of graphite or grapheneoxidein which a carbo-catalytic effect may apply (Navalon et al., 2014; Su and Loh, 2013).

#### f) Reaction GO and $GO-C_{60}$ with Cellulose

Nano cellulose – nano whiskers– can be prepared by hydrolysis of cellulose polymer (Bode son et al., 2006). Optimum conditions for the preparation depend on concentration of the employed acid (H<sub>2</sub>SO<sub>4</sub>HCl), ratio of cellulose and acid, time of hydrolysis and on reaction temperature. Naturally, the result is also influenced by the type of the initial cellulose which can come from hard or soft wood, bamboo, sisal, cotton etc. All those factors in fluenceyield and size of the prepared cellulose nano fibers (loelovich, 2012; Li and Rage uskas, 2011). The prepared nano crystalline cellulose can bechemically modified, e.g. esterifed, carboxy lated oroxidized (Peng et al., 2011). It may be also used asa composite inpolymersor as a matrix for metal nano particles.

In our case we performed hydrolysis of methyl car boxy cellulose in presence of suspensions of GO and  $\text{GO-C}_{60}$ . We assumed that mutual interconnection

may occur by a chemical reaction (e.g esterification, interconnection with C-O-C bond etc.) or physic chemical bond (e.g. hydrogenbonds). The foils prepared by vacuum filtration of the reaction product were subject to microscopic surface analysis, FT-IR, TGA and DSC analysis.

Morphology of the surfaces as shown by microscopic analysis (see Fig. 12) was different and it suggested potential method of interconnection between GO /  $GO-C_{60}$  and nanocellulose. The detailed morphology (electron microscope) of the prepared composite foils is shown in Fig.13 and 14.



*Figure 12*: Roughmorphology of the foil surfaces after the mutual action of  $GO-C_{60}$  (a) and GO (b) with cellulose







*Figure 14* : SEM of GO-cellulose foils: scale10  $\mu$ m and 20  $\mu$ m

# g) FT-IR Analysis of a Product of GO and $GO_{60}$ Reaction with Cellulose (Nano cellulose)

We assume that in the first case the cellulose hydrolysis was only partial and whiskers of nano cellulose generated after the hydrolysis were of bigger size. At the same time, there was no reaction of C=O groups, unlike in the case of GO, where IR analysis of

the product of reaction with cellulose identified no vibration at 1722 cm<sup>-1</sup>, which was present in the spectrum of the original GO. The other vibrations characterizing the groups C-O, C=C, C-O and C-O-C shifted their frequencies and also the mutual absorption ratios were different (Fig.15).

The same applies in respect to the initial GO, GO-C<sub>60</sub> and cellulose (compare with the spectrums in Fig.4)





*Figure 15* : IR spectrums of products of reaction of  $GO-C_{60}$  (a) and GO (b) with cellulose and (c) the initial cellulose alone

For the new products in both cases the IR spectrums did not contain the peaksat1278 cm<sup>-1</sup> (1274 cm<sup>-1</sup>) and 1228 cm<sup>-1</sup> which in the original spectrums  $GO-C_{60}$  and GO had the assigned vibrations of epoxy groups.

Similar results, i.e. demonstration of deoxidizing (reduction) process, were described for the mutual reaction of GO with heparin (Wang et al., 2012), with a solution of cellulose in 1-butyl-3-methylimidazolium chloride (Peng et al., 2012) and with chi to san– starch (Rodrigues-Gonzales et al., 2012).

#### h) Thermal Stability of Products of GO and $GO_{60}$ Reaction with Cellulose(Nano cellulose)

TGA curves of samples (see Fig. 16) of composites can be divided into several sections with

different slopes, i.e. different weight loss rates. This division, including corresponding temperature intervals and corresponding weight losses, is shown in Tables 4 and 5, which provide parameters of the detected thermal processes on the DSC curve.

For composite samples the DSC curve showed one peak corresponding to an endothermic process and two peaks corresponding to exothermic processes. The second exothermic process was very substantial in both the samples. For the GO-C<sub>60</sub>- cellulose sample the exothermic process started at 319.8°C and the peak area on the DSC curve was 3379.2 kJ/kg. Equally significant exothermic process in the GO-cellulose sample started at 341.5°C and the peak area on the DSC curve was 5261.4 kJ/kg.



a) GO-C<sub>60</sub>- cellulose (degradation medium: air, air flow rate 20 ml/min, temperature 25-600° C, heating rate  $10^{\circ}$ /min, sample weight 9.0 mg)



b) GO - cellulose (degradation medium: air, air flow rate 20 ml/min, temperature 25-600° C, heating rate10°/min, sample weight 9.0 mg)



*Figure 16 :* IR spectrum of composites GO-C<sub>60</sub>-cellulose a), GO-cellulose b), initial cellulose c) (degradation medium: air, air flow rate 20 ml/min, temperature 25-600° C, heating rate10°/min, sample weight 11.79 mg)

Sample No.	Interval No.	Temperature range (°C)	Weight loss (%)
	1	25.0 - 42.4	0.3
	2	42.4 - 123.9	6.9
	3	123.9 – 168.9	11.9
$GO-C_{60} - Cel.$	4	168.9 – 347.8	18.5
	5	347.8 - 474.3	36.4
	6	474.3 - 600.0	6.8
	1	25.0 - 57.3	0.9
	2	57.3 – 120.8	10.1
	3	120.8 – 144.6	3.0
GO – cel.	4	144.6 – 180.6	13.2
	5	180.6 – 396.5	20.6
	6	396.5 - 522.6	48.3
	7	522.6 - 545.0	2.5
	1	25.0 - 62.9	8.5
	2	62.9 - 124.9	55.6
cellulose	3	124.9 – 265.8	3.3
	4	265.8 - 333.4	15.8
	5	333.4 - 600.0	10.5

*Table 4* : Division of TGA curves into temperature intervals

Table 5 : Parameters of the ongoing thermal processes (DSC)

Sample No.	Thermal process No.	Temperature range (°C)	∆H (kJ/kg) *	H <sub>f1</sub> (mW)	ΣΔH (kj/kg)
	1	31.6 – 115.3	159.9	5.7	
GO-C <sub>60</sub> – cel.	2	115.3 – 187.2	-251.1	18.7	-3470.2
	3	319.8 – 531.0	-3379.0	59.2	
	1	25.0 – 133.1	757.5	17.7	
GO – cel.	2	133.1 – 208.1	-581.2	41.7	-5085.1
	3	341.54 – 557.2	-5261.4	76.8	
	1	25.0 – 154.0	3053.7	86.2	
cellulose	2	247.9 - 344.8	-211.3	12.2	+2805.7
	3	372.0 - 414.6	-36.7	2.3	

 $^{*}\Delta H =$  thermal effect of the process based on DSC curves

 $(\Delta H > 0...endothermic process, \Delta H < 0...exothermic process)$ 

The comparison of thermal stabilities of the prepared GO and GO-C60composites with cellulose indicates that the foil prepared from GO is thermally more stable but its decomposition releases more thermal energy.

A principle difference can be found when we compare thermal stability (weight loss) of the initial foils of GO and GO-C60 and thermal stability of their composites with cellulose. The weight loss in the temperature interval 25-220°C was 63% for GO foil sand 72% for  $GO-C_{60}$  foils. For the composites the weight loss was 32% and 22% respectively, which means a major difference. A completely different is the thermal decomposition of the initial cellulose. The decomposition in the temperature interval 25-154°C is accompanied by an endothermic process with the thermal effect 3053.7 kJ/kg and the weight loss of the sample is 64%.

# i) GO Reaction with B- 1,3-1,6- D-Glucan

Glucans rank among homopolymsacharides, they have a long chain with only one structural component-glucose (hexose). Glucose in the chain is attached in the positions 1,3 and 1,6. Smaller chains branch off from the main chain. The structure of glucans is of extraordinary importance inactivation of the immunity system where branching of the side chains plays a majorrole.

β-D-glucans are indigestible polysaccharides that occur in nature in sources such as cereal grains, yeasts, fungi, bacteria and algae. Biological effects of beta-glucans are manifested at several levels. The main role consists in activation of immunity system cells (macrophage) and they also performanti-carcinogenic, antimicrobial, antiviral and antialergic activities. Betaglucans also have a radio protective effect--they deactivate free radicals (Chovancová and Šturdík, 2005).

We performed reaction of beta-glucan with graphene-oxide under ultrasonification conditions with out sulfuric acid and in acid environment.

# j) FT-IR Analysis of the Foils (GO-BG and GO-BGH+)

The obtained spectrums of the products are shown in Fig.17from left to right: GO- $\beta$ G, GO- $\beta$ GH+ and initial  $\beta$ G. In the last picture d) the three spectrums are compared.





d) Combined spectrums. a) black + b) green + c) red

*Figure 17 :* IR spectrums of the reaction products: a)GO-BG , b) GO- BGH+ ,c) BG, d) comparison of the spectrums shown ina + b + c

At the first sight the IR spectrums shown above are similarbuta more detailed inspection of the GO-BG and GO- BGH + spectrums shows differences in absorbance values, mutual ratio of peak intensity and in shifts of frequencies of decisive bond vibrations, e.g.:

	GO-BG	GO- ßGH⁺
C-O-C	1018 cm <sup>-1</sup>	1044 cm <sup>-1</sup>
-OH	3255 cm <sup>-1</sup>	3286 cm <sup>-1</sup>
-C=C-	1620 cm <sup>-1</sup>	1631 cm <sup>-1</sup>

The values of the initial GO are: C-O-C 1068 cm<sup>-1</sup>, 9.79 cm<sup>-1</sup>, -OH 3149-3186 cm<sup>-1</sup>, -C=C- 1613 cm<sup>-1</sup>.

#### k) Thermal Stability of the Product (GO-BG, GO-BGH+) and Initial BG

The measured results in a graphic form are provided in Fig.18-20 and interpretation of the TGA and DSC curves is provided in Tables 6-7.The DSC curve of BG features one peak corresponding to an end other micthermal process with minimum weight loss and prominent, partly overlapping peaks that correspond to exothermic thermal processes which start at 257°C and are accompanied by a significant loss of the sample weight.

A common characteristic of both the prepared composite products is that their weight loss curves (TGA)can be approximated with a line –continuous linear reduction of weight, unlike the step weight loss in case of  $\beta$ G which was up to 70% (see Fig.18) and in case of GO up to 60% (see Fig.5). The products differ from each

other by the number of exothermic effect sand the shift of the last exothermic effect by  $42^{\circ}$ C in favor of GO-B G H+.

The overall thermal effect in the course of decomposition decreases from the initial BG to the GO-BGH+ composite (see Tab.2).



*Figure 18 :* Thermal analysis of BG (degradation, medium: air, air flow rate 20 ml/min, temperature 25-600°C, heating rate10°/min, sample weight 11.03 mg)



*Figure 19 :* Thermal analysis of GO-BG (degradation medium: air, air flow rate 20 ml/min, temperature 25-600°C, heating rate10°/min, sample weight 11.47 mg)



*Figure 20 :* Thermal analysis of GO-BGH+ (degradation medium: air, air flow rate 20 ml/min, temperature 25-600°C, heating rate10°/min, sample weight 10.77 mg)

Sample No.	IntervalNo.	Temperature range (°C)	Weight loss(%)
	1	25.0 - 114.4	5.5
	2	114.4 – 278.1	8.6
80	3	278.1 – 330.3	52.2
DG	4	330.3 - 412.7	17.0
	5	412.7 – 462.4	14.8
	6	462.4 - 475.9	14.8
	1	25.0 - 149.4	9.9
	2	149.4 – 225.5	12.3
	3	225.5 – 292.3	4.7
GO- 11G	4	292.3 - 337.9	26.7
	5	337.9 - 438.4	15.7
	6	438.4 - 529.8	30.7
	1	25.0 – 124.2	9.9
	2	124.2 - 204.2	4.2
	3	204.2 - 230.8	10.9
GO-DGH	4	230.8 - 321.6	22.5
	5	321.6 - 480.9	25.9
	6	480.9 – 550.2	26.9

#### Table 6: Division of the TGA curves into temperature intervals \*

<sup>x</sup>-the indicated intersections of tangents to the respective bends of the TGA curve

Table 7: Parameters of the thermal processes (DSC)

Sample No.	Thermal processNo.	Temperature range (°C)	∆H (kJ/kg) *	H <sub>f1</sub> (mW)	ΣΔH (kj/kg)
00	1	25.0 – 153.9	805.9	21.3	
bG	2	257.5 - 528.4	-4392.5	81.0	-3586.6
GO- ßG	1	25.0 – 157.0	825.3	16.9	
	2	157.0 – 228.8	-147.1	8.9	-2734.9
	3	311.0 – 356.5	-36.3	6.2	
	4	383.3 – 539.3	-3376.8	66.4	
GO- βGH⁺	1	25.0 -145.9	685.2	16.7	
	2	197.2 – 250.0	-313.5	38.2	-2084.7
	3	425.4 - 589.3	-2456.4	46.2	

 $\Delta H$  = thermal effect of the process based on DSC curves ( $\Delta H$  > 0...end other micprocess,  $\Delta H \square 0 \cdots e_X$  other micprocess) Hfl =height of the peak of athermal processon the DSC curve in an absolute value related to the point corresponding to the beginning of the thermal process

# III. Conclusion

Joint oxidation of graphite and fullerene  $C_{60}$  in the ratio2:1makes it possible to prepare a compound din form of compact foils but it has a lower thermal stability and its thermal decomposition is accompanied by thermal effects that are 30% bigger than effects produced by foils without fullerene. A method of attachment between oxidized graphite and fullerene has not been demonstrated. The prepared foil can also have other chemical and physicochemical properties thanks to the fullerene molecule (Trosh in et al., 2008).

Thermal decomposition of composites of GO and GO-C<sub>60</sub> with nano cellulose is accompanied by bigger thermal effects than the effects produced by the initial GO and GOC<sub>60</sub> foils. The same applies also for  $\beta$ -glucan in respect to GO. However, the thermal effect of the decomposition is lower than for  $\beta$ -glucan alone.

The products we prepared were in the form of foils (membranes), with the exception of the "blank test" of fullerene oxidation. GO-foils (papers) are prepared by

vacuum filtration of GO dispersion. This is the basic method of its preparation which we have also applied. We are fully aware of the fact that mechanical, electronic, chemical and biological properties and the related toxicological properties are affected by many factors which may have influenced our results of thermal stability measurements. The first factor is the method of GO preparation and thus there sultingratio of C/O and topology of the C-skeleton (its defects). The composition of the liquid phase and the concentration (Park et al., 2012) of the filtered suspension influence the foil thickness, as well as the filtration rate, and if the suspension is ultrasonificated then also its duration, temperature and power of the device play a role (Liao et al., 2011). Properties of foils are also influenced by some specific treatments, such as washing of foils with solution of MCl<sub>2</sub> (Ca, Ba, Mg), while the carbon layers are connected in a plane, across via dialdehyde (Hu et al., 2011), or its impregnatione.g. with Ti, Ag, Cu<sub>2</sub>O (Chen et. al., 2011), expansion of the inter layer space (Zhu et al., 2012) etc.

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